## $LiAl_yNi_{0.5-y}Mn_{1.5}O_4$ (y = 0.0–0.4) AS 5-VOLT CATHODES FOR LITHIUM BATTERIES

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One method adopted to surmount the poor cyclability problem of LiMn<sub>2</sub>O<sub>4</sub> is the partial substitution of the manganese with a metal ion that can stabilize the host structure during the charge-discharge processes. Although such substitutions can enhance the structural stability of the host structure, there is often a penalty in terms of the 4-volt capacity. This presentation reports the structural and lithium intercalation properties of LiAl<sub>y</sub>Ni<sub>0.5-y</sub>Mn<sub>1.5</sub>O<sub>4</sub> (y = 0.0, 0.1, 0.2, 0.3 and 0.4). The compositions were prepared by a solid-state fusion method from Li<sub>2</sub>CO<sub>3</sub>, MnO<sub>2</sub>, NiO and Al(OH)<sub>3</sub> at 800°C for 24 h in air.

The x-ray diffraction peaks recorded for the Alsubstituted compositions were indexable in the Fd3mspace group. A few minor peaks also appeared, especially at the higher Al concentrations, suggesting the presence of some extraneous phases like  $\gamma$ -LiAlO<sub>2</sub> [1]. According to Ohzuku et al. [2] the integrated intensity ratios of the (400)/(311) and (220)/(311) peaks are indices of the occupancy of the substituent ions in the 8*a* lithium sites. An analysis of our XRD data showed increased occupancy of the substituent Al in the 8*a* lithium lattice sites especially at Al stoichiometries above 0.3.

The various compositions were cycled between 3.30 and 4.95 V at a 0.1 C rate in 2032-type coin cells with lithium as the anode. They showed two regions of electrochemical activity: one in the 4-volt region and the other in the 5-volt region. For comparison, the cycling performance (between 3.00 and 4.20 V) of an unsubstituted LiMn<sub>2</sub>O<sub>4</sub> sample was also studied. Substituting the trivalent Al ion for Ni in LiNi0.5Mn1.5O4 should result in the reduction of an equivalent amount of  $Mn^{4\scriptscriptstyle +}$  to  $Mn^{3\scriptscriptstyle +}\!,$  yielding materials with increased 4-volt activity (Table 1). It can also be seen from Table 1 that the cumulative capacity of the spinel increases when the Ni content is increased from 0.1 to 0.5 atom/molecule. At the latter composition, theoretically, there is only one active species: the 5-volt-oxidizable Ni<sup>2+</sup> species, whose two-electron transfer oxidation to Ni<sup>4+</sup> should result in higher capacities. In fact, the first-cycle discharge capacity of  $LiNi_{0.5}Mn_{1.5}O_4$  was 114 mAh/g as against 108 mAh/g for LiMn<sub>2</sub>O<sub>4</sub>. A first-cycle capacity of 110 mAh/g was obtained with LiAl<sub>0.1</sub>Ni<sub>0.4</sub>Mn<sub>1.5</sub>O<sub>4</sub>. Higher capacities at higher voltages should translate into higher power densities for the Ni-substituted spinels.

The results of extended cycling studies (Fig 1) show that the unsubstituted spinel gave a cumulative first-cycle discharge capacity of 108 mAh/g, which faded to 90 mAh/g in the 25th cycle, registering charge retention of 84%. The corresponding values for  $\text{LiNi}_{0.5}\text{Mn}_{1.5}\text{O}_4$  were 114 and 110 mAh/g and 97%. Thus, substitution with Ni enhanced the first cycle capacity as well as the cyclability of the spinel. In the case of the Al-substituted compositions with y = 0.1, there was an initial surge in the capacity (110 mAh/g). However, as the y values were increased, the capacities of the Al-substituted materials dropped. The fall in capacity with an increase in the y value is a consequence of the increased propensity of these ions to occupy the 8a lithium sites as their concentration is increased. Their occupancy in the lithium sites can impede transport of the Li ions during the charge-discharge processes. This is reflected in the decreased charge retention of the Al-substituted materials (e.g., 71% over 60 cycles for y = 0.1). Thus, Al does not seem to lead to structural stability. It was also observed that the capacities of the Al-substituted materials registered a general increase from the first cycle to the  $10^{\text{th}}$ , followed by a slow decay in subsequent cycles.

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#### References

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Table 1. The theoretical and observed first-cycle	
capacities of the various LiAlyNi0.5-yMn1.5O4 compositions	;

Composition	4-volt capacity		5-volt capacity		
					C. C.
	T. C.	0. C.	T. C.	0. C.	
LiMn <sub>2</sub> O <sub>4</sub>	148.2	108.0			108.0
LiNi0.5Mn1.5O4	0.0	27.2	146.7	86.8	114.0
LiAl <sub>0.1</sub> Ni <sub>0.4</sub> Mn <sub>1.5</sub> O <sub>4</sub>	14.9	37.1	119.4	72.9	110.0
LiAl0.2Ni0.3Mn1.5O4	30.4	48.8	91.2	50.2	99.0
LiAl <sub>0.3</sub> Ni <sub>0.2</sub> Mn <sub>1.5</sub> O <sub>4</sub>	46.4	58.9	61.9	38.1	97.0
LiAl <sub>0.4</sub> Ni <sub>0.1</sub> Mn <sub>1.5</sub> O <sub>4</sub>	63.1	69.3	31.5	21.7	91.0

T. C. : Theoretical Capacity (mAh/g)

O. C. : Observed Capacity (mAh/g)

C. C. : Cumulative Capacity Observed (mAh/g)



Fig. 1. Cycling behavior of the various doped systems