The activation energy of self-discharge in anode and cathode for lithium-ion batteries Rachid Yazami^{a,b}

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Self-discharge is a common phenomenon in primary and secondary batteries. It results mainly from the thermodynamically predicted reactions between the electrolyte and the active electrode materials. As a matter of fact, most batteries in their charge state yield an open circuit voltage than can be wider than the stability window of the electrolyte. Therefore, the charge retention in a battery is mainly governed by the kinetics of the electrode/electrolyte reactions. The latter are usually thermally activated, which mean that the rate of selfdischarge will increase with temperature.

Most studies on the self-discharge in lithium-ion batteries were conducted on a complete cell, usually based on the $\text{Li}_x\text{C}_6/\text{LiPF}_6$ organic carbonate/ Li_yCoO_2 chemistry. However, since both Li_xC_6 and Li_yCoO_2 are metastable versus the electrolyte, their kinetics of self-discharge may involve different mechanisms due to differences in their chemical potential. A separate study in half-cells will allow distinguishing between contributions each one of the electrodes to the total self-discharge and to the reversible and irreversible capacity losses.

1/The anode side: Li/1M LiPF₆-EC(1):DMC(1)/graphite half-cells were cycled at ambient temperature under C/5 rate between 1.5 and 0V [1]. They were stored at different temperature between 55 and 80C for several days at the initial fully lithiated state LiC₆. The open-circuit voltage (OCV) was monitored during the aging. The remaining capacity was determined during the first charge (delithiation) and the new reversible capacity was accessed by galvanostatic cycling at the ambient temperature.

The time dependence of the OCV was found according to the following phenomenological law:

$$E(T,t) = E^{0}(T) + \frac{RT}{F} \ln \frac{1}{(1-x_{\infty})e^{-kt} + x_{\infty}} + \chi(T,t) \quad (1)$$

Where $\chi(T, t) = \chi_0(T) + A(T)t + B(T)t^2 + ... (2)$

The time constant k was found to follow the Arrhenius law and the corresponding activation energy was determined to be 48.8 kJ.Li-mole⁻¹. On the other hand, the total capacity loss was found to follow a first order kinetics law of the type:

$$Q(T,t) = Q_0 \exp(-\lambda(T).t)$$
(3)

where the time constant $\lambda(T)$ has an activation energy of 50.2kJ.(Li-mole)⁻¹ close to the one of k in eqn. (1). This suggests that the capacity loss and the voltage change have similar mechanism, which we attributed to lithium reaction with the electrolyte, which ultimately transform to the solid-state interphase (SIE) that builds up on the graphite surface.

2/The cathode side: For practical reasons, we used 1M LiClO₄ electrolyte solutions in the Li/LiCoO₂ half-cells for the self-discharge study. Similar cycling was applied between 3 and 4.2 V at ambient temperature, and then cells in their charged state (Li_{0.5}CoO₂, 4.2 V initial) were maintained at constant temperature between 55C and 75C for up to 6 weeks [2]. Here the OCV curves followed different time dependence:

$$E(t, T) = E(0, T) - K(T)t^{1/2}$$
, (4)

K(T) was found to obey the Arrhenius law with 41.3 $kJ(Li-mole)^{-1}$ as activation energy.

The relative capacity total loss 'x' (in %) fitted with the following time equation:

$$t = ax^2 + bx + t_0$$
, (5)

where the 'a' constant was found to be thermally activated with $81.20 \text{ kJ}(\text{Li-mole})^{-1}$ as activation energy. The differences in the activation energy from the OCV and the capacity loss measurement suggest the corresponding kinetics governed by different mechanisms. As a matter of fact, the OCV related to the Li-diffusion from the electrolyte to the electrode bulk structure, while the capacity loss was governed by surface reactions and phase transformations.

This study highlights the usefulness of the half-cell configuration in the kinetics of self-discharge in lithium-ion batteries.

References:

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