# Study on the Li de-intercalate mechanism of the $LiNi_{1/2}Mn_{1/2}O_2$ -based materials

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#### **INTRODUCTION**

LiNi1/2Mn1/2O2-based materials are one of the promising cathode materials of lithium secondary battery.  $LiNi_{1/2}Mn_{1/2}O_2$ -based materials adopt hexagonal unit cells like LiCoO<sub>2</sub> and LiNiO<sub>2</sub> and LiNi<sub>1/3</sub>Mn<sub>1/3</sub>Co<sub>1/3</sub>O<sub>2</sub> displays a reversible capacity of 170 mAh/g in the voltage range 2.5 to 4.5 V [1]. In addition, these materials show the superior characteristics of a larger capacity than LiMn<sub>2</sub>O<sub>4</sub> and better thermal stability than LiNiO2. On the other hand, the structure and physical properties of the Li1.  $_{v}Ni_{1/2}Mn_{1/2}O_{2}$ -based materials are still ambiguous. Detailed information on the structure and the valence state of the cations in  $Li_{1-\nu}Ni_{1/2}Mn_{1/2}O_2$ -based one is very important in order to improve the electrochemical properties of LiNi1/2Mn1/2O2-based material and, therefore, the relationships between the chemical composition, the structure, and electrochemical properties were studied in this study.

## **EXPERIMENTAL**

Li<sub>1-y</sub>Ni<sub>1/2</sub>Mn<sub>1/2</sub>O<sub>2</sub>-based materials were synthesized in air at 1223-1273 K for 12-24 h and de-lithiated samples were electrochemically prepared using coin-type cells with Li/1M LiPF<sub>6</sub> in EC:DEC(1:1)/samples. The M (M =Ni, Co, Mn, Ti) valence states of samples were determined by the M K-edge XANES spectra. Structure and physical property changes were investigated by synchrotron X-ray diffraction ( $\lambda = 0.5$  Å), neutron diffraction, M K-edge EXAFS measurements.

## **RESULTS AND DISCUSSION**

 $LiNi_{1/2}Mn_{1/2}O_2$  was a single-phase and adopted the  $\alpha$ -NaFeO<sub>2</sub> structure. Structural analysis using synchrotron and neutron diffraction data demonstrated that the lattice parameters of  $LiNi_{0.5}Mn_{0.5}O_2$  are a = 2.892Å and c = 14.302 Å and that the chemical composition can be expressed referring to the Wyckoff positions 3aand 3b as  $[Li_{0.91}Ni_{0.09}]_{3a}[Li_{0.09}Mn_{0.5}Ni_{0.41}]_{3b}O_2$ . In charging process, the structure of charged Li1- $_{v}Mn_{0.5}Ni_{0.5}O_{2}$  (y = 0.5) was determined, corresponding to the composition for showing rechargeable capacity of ca. 140 mAh/g. The results show that divalent nickel metal is oxidized to trivalent after charging, in association with the phase transition from hexagonal (R3m) to monoclinic (C2/m) symmetry [2]. The crystal structures and electron density distributions of the layered oxide Li1- $_{y}Ni_{0.5}Mn_{0.5}O_{2}$  (y = 0.5) was studied using a combination of Rietveld analysis of high-resolution synchrotron powder X-ray diffraction data and the maximum entropy method (MEM). Structural analysis revealed that  $Li_{1}$ .  $_{\rm v}{\rm Ni}_{0.5}{\rm Mn}_{0.5}{\rm O}_2$  (y = 0.5) has the lattice parameters a =

4.934 Å, b = 2.852 Å, c = 5.090 Å,  $\beta = 108.8^{\circ}$  and adopts the space group C2/m. The chemical formula can be expressed as  $[Ni_{0.0815}]_{2a} \{Li_{0.5}Ni_{0.0115}\}_{4i} [Mn_{0.5}Ni_{0.407}\square_{0.093}]_{2d}$ O<sub>2</sub> [3]. The electron density map obtained using MEM clearly shows that most of the Li ions migrate from the octahedral 2*a* site to the tetrahedral 4*i* site during Li deintercalation.

 $LiNi_{1/2}Mn_{2/5}Ti_{1/10}O_2$  and  $LiNi_{1/3}Mn_{1/3}Co_{1/3}O_2$  were also single-phase and adopted the  $\alpha$ -NaFeO<sub>2</sub> structure. Structural analysis using synchrotron and neutron diffraction data demonstrated that the lattice parameters of LiNi<sub>1/2</sub>Mn<sub>2/5</sub>Ti<sub>1/10</sub>O<sub>2</sub> are a = 2.895 Å and c = 14.317 Å and that the chemical composition can be expressed referring to the Wyckoff positions 3a and 3b as  $[Li_{0.88}Ni_{0.09}Ti_{0.03}]_{3a}[Li_{0.12}Ni_{0.41}Mn_{0.40}Ti_{0.07}]_{3b}O_2$  and that the lattice parameters of  $\text{LiNi}_{1/3}\text{Mn}_{1/3}\text{Co}_{1/3}\text{O}_2$  are a =2.860 Å and c = 14.22 Å and that the chemical composition can be expressed referring to the Wyckoff positions 3a and 3b with space group R3m as  $[Li_{0.97}Ni_{0.03}]_{3a}[Li_{0.03}Mn_{0.33}Ni_{0.30}Co_{0.33}]_{3b}O_{2}$  [4]. The Li deintercalate mechanism of these materials will be compared with  $LiNi_{0.5}Mn_{0.5}O_2$  and, furthermore, the relationships between the structure changes and their electrochemical properties will be discussed.

#### References

- 1 N. Yabuuchi and T. Ohzuku, *J. Power Sources*, *119-121*, (2003) 171-174.
- 2 Y. Arachi, H. Kobayashi, S. Emura, Y. Nakata, M. Tanaka, and T. Asai, *Chemistry Letters 32* (2003) 60-61.
- 3 H. Kobayashi, H. Sakaebe, H. Kageyama, K. Tatsumi, Y. Arachi, and T. Kamiyama, J. Mater. Chem., 13 (2003) 590-595.
- 4 H. Kobayashi, Y. Arachi, K. Tatsumi, and H. Kageyama, J. Mater. Chem., 14 (2004) 40-42.