

Characterization of the $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ system obtained by lithium ion-exchange

C. Delmas¹, F. Tournadre¹, S. Komaba^{1,3}, M. Ménétrier¹,
I. Saadoun¹, P. Willmann², and L. Croguennec¹

¹ ICMCB-CNRS and ENSCPB, Université Bordeaux I
87 Av. Dr A. Schweitzer, 33608 Pessac cedex, France

² Centre National d'Etudes Spatiales,
18 Av. Edouard Belin, 31401 Toulouse cedex 4, France

³ Present address: Iwate University, Morioka,
Iwate 020-8551, Japan

Layered LiMO_2 materials obtained by lithium ion exchange in P2 sodium containing phases demonstrated interesting properties as positive electrode materials for Li-ion batteries.

Recently, the crystal structure analysis and battery performance of $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ obtained by Li/Na ionic exchange in molten salts were studied [1, 2]. As shown in Fig. 1, three tetrahedral surroundings are available for lithium in the interslab space of the $T^{\#2}$ stacking; but for $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ it was determined from nuclear density Fourier difference map calculations that lithium ions are only distributed among the 8e and $8f_{\text{edges}}$ sites, with a larger occupancy of the latter. $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ shows good electrochemical performances, with several phase transitions and an irreversible transition towards an O6-type phase at the end of the 1st cycle (Fig. 2).

The redox processes were characterized from electrical properties as well as XANES and ⁷Li MAS NMR. Upon lithium deintercalation, cobalt oxidation is observed, whereas upon lithium intercalation manganese and cobalt reduction seems to occur even if for the latter it was not expected.

The phase transition from $T^{\#2}$ to O6 stacking observed for $\text{Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ upon increasing temperature (above 300°C) was studied using *in situ* X-ray diffraction (Fig. 3). The kinetics study of this phase transformation is still in progress. This $\text{O6}_{\text{HT}}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ phase is stable up to 480°C before a decomposition into Li_2MnO_3 and Co_3O_4 at higher temperature. Note that $\text{O6}_{\text{HT}}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ exhibits comparable battery performances with the $T^{\#2}$ ones.

Acknowledgements:

We thanks M. Morcrette (LRCS, Amiens, France) for the *in situ* diffraction experiments and preparation of the samples for the XAFS experiments, E. Suard (ILL, Grenoble, France) for the neutron diffraction experiments and, Y. Shao-Horn and F. Weill (ICMCB, Bordeaux, France) for the microscopy experiments. This work was financially supported by CNES and Région Aquitaine (CPER Véhicule Electrique 21-13). One of us (S.K.) thanks the New Energy and Industrial Technology Development Organization (NEDO) for financial support of Industrial Technology Research Grant Program.

References

- [1] F. Tournadre et al., *Chem. Mater.* **16**, 1411 (2004)
[2] F. Tournadre et al., *Chem. Mater.* **16**, 1418 (2004)

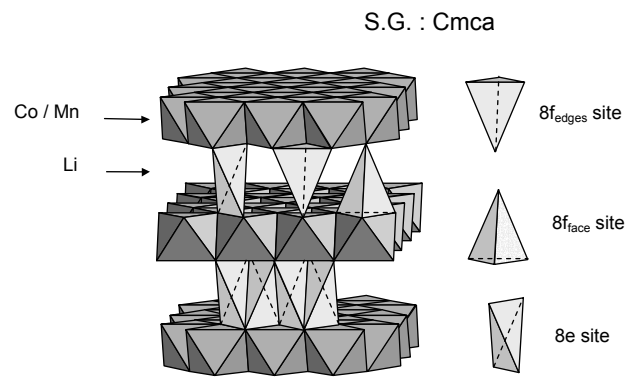


Figure 1: Schematic structure of $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$. Lithium ions are only distributed in the 8e and $8f_{\text{edges}}$ tetrahedral sites.

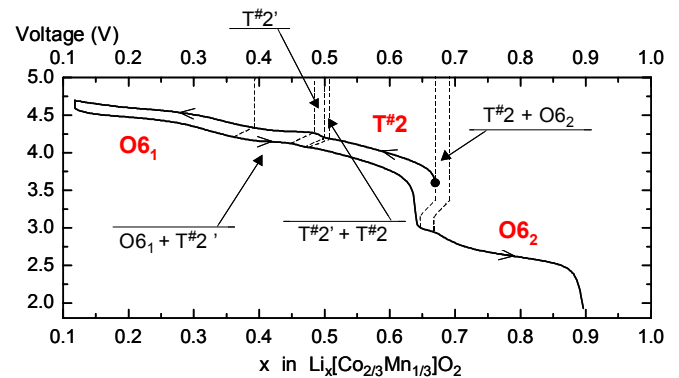


Figure 2: Phase diagram of the $\text{Li}_x\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ system reported on the galvanostatic curve (1st charge and discharge) of an $\text{Li}/T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ cell.

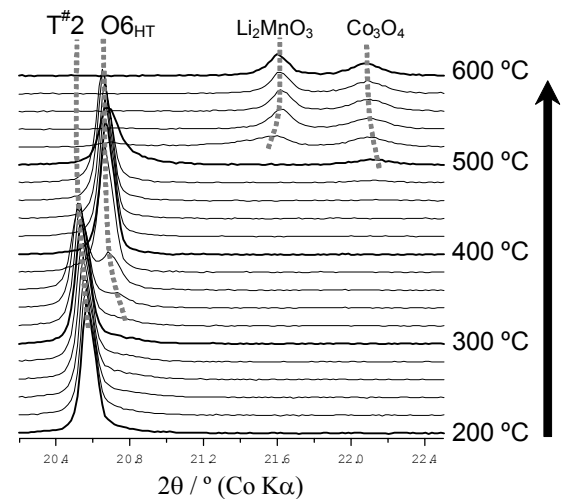


Figure 3: Thermal (*in-situ*) powder XRD patterns measured during heating process of $T^{\#2}\text{-Li}_{2/3}\text{Co}_{2/3}\text{Mn}_{1/3}\text{O}_2$ up to 600 °C in air. Each pattern has been recorded at a stabilized temperature.