COMPARISON OF QUANTUM WELL INTERMIXING IN GaAs STRUCTURES USING A LOW TEMPERATURE GROWN EPITAXIAL LAYER OR A SiO₂ CAP

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The mechanism of quantum well intermixing (QWI) involves the creation of a non-equilibrium concentration of point defects which diffuse through the quantum well (QW) and adjacent barrier layers during a high temperature anneal treatment. This causes inter-diffusion and consequent change in the QW composition and thickness [1] leading to changes in the effective bandgap of the QWs which shows up as a blue-shift in the room temperature photoluminescence (PL). Then by spatially defining the areas of modified regions across a wafer, this QWI technique has been widely proposed as a method for implementing photonic integration [2]. This work reports on a comparison of the use of a low temperature grown epitaxial InGaP (LT-InGaP) or PECVD deposited SiO2 with rapid thermal annealing (RTA) for 60s at temperatures up to 900°C to induce QWI. Samples consisted of single QW structures in which the wells and barriers had either changes in the group III (InGaAs/GaAs) or group V (InGaAs/InGaAsP) compositions only. This allows us to separate out the defects migrating as group III or group V species. The results in fig.1 show that the LT-InGaP cap most strongly affects the InGaAs/ InGaAsP sample by inducing a larger bandgap blueshift, but there is a much reduced photoluminescence (PL) signal. This reduced intensity we ascribe to lateral composition modulation which is greatly increased by the RTA treatment, as shown in fig.2. For the InGaAs/GaAs samples the blue-shift after RTA is similar for both capping layers as seen in fig.3; also, the PL intensities remain strong for anneals up to 900°C. These results will be explained based on previous explanations for the SiO_2 cap behaviour [1] and for the defects hypothesized to be present in LT-InGaP based on our previous work using LT-InP to induce QWI in InP-based QW structures [3].

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[2] J. H. Marsh, S. I. Hansen, A. C. Bryce, R. M. De La Rue, Optical and Quantum Electron. <u>23</u>, S941 (1991).

[3] B. E. Gordon, A. S. W. Lee, D. A. Thompson, B. J. Robinson, Semicond. Sci. Technol. <u>18</u>, 782 (2003).



Figure 1. Peak PL wavelength after anneal for the variously capped InGaAs/InGaAsP samples.



Figure 2. Cross-sectional TEM for samples with InGaAs QW/InGaAsP Barriers. (a) as-grown, and (b) after anneal at $925^{\circ}C$.



Figure 3. Peak PL wavelength after anneal for the variously capped InGaAs/GaAs samples.