

LITHIUM INSERTION INTO RAMSDELLITE

$\text{Li}_2\text{Ti}_3\text{O}_7$

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Introduction

Lithium-ion batteries have been emerging as power sources for modern electronics because they have high working voltage, high energy density, and good cyclability [1, 2]. In these batteries, numerous effort have been made to find alternative anode materials to replace graphite in Li-ion batteries. Notably, lithium titanium oxides due to an average voltage around 1.5V, such as the spinel phase $\text{Li}_4\text{Ti}_5\text{O}_{12}$ [3, 4] or the ramsdellite phase $\text{Li}_2\text{Ti}_3\text{O}_7$ [5, 6] have been proposed. We have already reported the synthesis of a mixed lithium titanium spinel $\text{Li}_4\text{Ti}_5\text{O}_{12}$ obtained via a sol-gel process in non aqueous media [4]. In this paper, we report the main electrochemical features achieved for the ramsdellite $\text{Li}_2\text{Ti}_3\text{O}_7$ synthesized via a sol-gel process.

Experimental

The precursor solution for $\text{Li}_2\text{Ti}_3\text{O}_7$ was a mixture of 0,033 mole of titanium alkoxide, $\text{Ti}(\text{OiPr})_4$ and 0.0264 mol of lithium acetate dihydrate ($\text{LiOAc} \cdot 2\text{H}_2\text{O}$). Hydrolysis, generated in situ by the dissolution of the hydrated lithium salt, followed by polymerisation occurs and a white monolithic gel is obtained after 1H. After calcination of the gel at 1050°C during 24H and quenching, the sample was characterized by X-ray diffraction.

Results and discussion

The powder X-ray diffraction data of the sample is shown in Fig.1. All the diffraction peaks are indexed using an orthorhombic cell with the following parameters $a = 5.045 \text{ \AA}$, $b = 9.542 \text{ \AA}$ and $c = 2.943 \text{ \AA}$ corresponding to the ramsdellite structure $\text{Li}_2\text{Ti}_3\text{O}_7$.

Fig.2 displays the chronopotentiometric behaviour of $\text{Li}_2\text{Ti}_3\text{O}_7$ at constant current density between 3V-1V in 1M LiClO_4/PC . At C/20, two insertion process appear. The first step involves a faradaic yield of 0.6 Faraday per mole with a decrease in potential from 2.9V up to 1.5V. The second step with a lower slope located around 1.4V corresponds to the accomodation of 1.1 additional lithium ion. During cycling experiments performed at discharge charge rate of C/20, a stabilization of the specific capacity around 120 $\text{Ah} \cdot \text{kg}^{-1}$ occurs after the 35th cycle (Fig.3).

Reference

4. S.Bach, J.P. Pereira-Ramos and N.Baffier, *J.Power Sources*, 1999, 81-82, 273.
5. M.E.Arroyo, E.Moran, A.Varez and F.Garcia-Alvarado, *Mat.Res.Bull.*, 1997, 32, 993.
6. S.Garnier, C.Bohnke, O.Bohnke and J.L.Fourquet, *Solid State Ionics*, 1996, 83, 323.

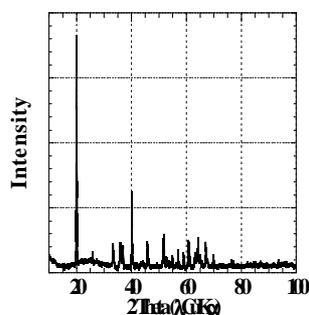


Fig.1 : Powder X-ray diffraction pattern of $\text{Li}_2\text{Ti}_3\text{O}_7$ obtained at 1050°C/24H after quenching.

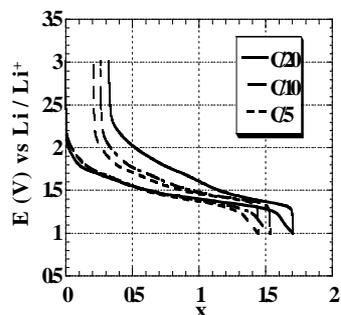


Fig.2 : Chronopotentiometric behaviour of $\text{Li}_2\text{Ti}_3\text{O}_7$ in 1M LiClO_4/PC .

1. B.Scrosati, J.Electrochem.Soc., 1992, 139, 2776
2. T.Ohzuku, in Lithium Batteries. New materials, developments and Perspectives, ed. G. Pistoia, Elsevier, Amsterdam, 1994, p.239.
3. E.Ferg, R.J.Gummow and A. de Kock, J.Electrochem.Soc., 1994, 141, L147

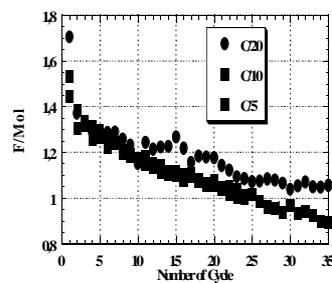


Fig 3 : Evolution of the faradaic yield as a function of the number of cycles for different discharge-charge rates.